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Stress-enhanced Cu-to-Cu Bonding for MEMS Packaging

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To replace Al–Ge eutectic bonding, low-temperature direct Cu-to-Cu bonding was developed in this study. Lattice distortion and thus a hardened Cu subsurface conducted by air plasma bombardment, instead of surface activation, contributed to a compressive residual stress component and an accelerated Cu atom diffusion. This gave rise to a significant improvement in direct Cu bonding strength. Subjected to 3-min plasma exposure and the following deoxidation treatment using catalyzed formic acid vapor, robust Cu-to-Cu bonding with the joint strength up to 31.7 MPa can be achieved when bonded at 250 $^{\circ}$ C for 5 min under a loading pressure of 10 MPa in N₂.

1. Introduction

Al–Ge eutectic bonding has been developed by InvenSense for inertial measurement unit (IMU) sealing.^(1,2) The Al–Ge bonding process needs to be performed at 450 °C or above, and the bonding period is at least 70 min. To achieve IMU sealing at low temperatures within a short time, in this study, we proposed a modified direct Cu bonding with no high vacuum. The process temperature is aimed to be 250 °C or even below. Bonding time can be decreased from 70 to 5 min. Complicated thin film deposition can thus be omitted.

Traditional direct Cu bonding is usually performed at temperatures above 350 °C under high vacuum (~ 10^{-3} Torr).⁽³⁾ A postannealing is necessary to eliminate the bonding interface and thus increase joint strength.⁽⁴⁾ Recently, modified processes to enhance direct Cu bonding have been suggested.^(5–11) Fujitsu Laboratories proposed a surface grain refining pretreatment, which combines diamond cutting, organic acid vapor, and vacuum ultraviolet treatments.^(12,13) Diamond cutting is performed to acquire a fine crystal layer underneath the surface, and formic acid vapor removes Cu oxides on the cut surface. A subsequent vacuum ultraviolet irradiation dissociates residual formate. Through this combined process, successful bonding can be

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carried out at 175 °C. It was proposed that fine grains underneath the diamond bit-cut surface accelerate Cu atom interdiffusion between two faying faces.

On the basis of the above surface modification concept, in this study, we combine air plasma bombardment with a following catalyzed formic acid vapor treatment, which aims to achieve robust Cu-to-Cu direct joints at 250 °C or below for the application of MEMS sealing. Instead of surface activation, the low-vacuum air plasma bombardment conducted is to control the surface physical characteristics of the faying face, and formic acid vapor catalyzed by Pt is adopted for deoxidation prior to thermal compression bonding.

2. Experimental Procedures

Squared silicon chips with $2.1 \times 2.1 \text{ mm}^2$ dimensions and 550-nm-thick sputtered Cu films on the Si chips were prepared. The Cu patterns were of square shape for which the width of the sealing frames was 136 nm. Low-vacuum air plasma bombardment and formic acid vapor treatment were carried out in sequence before bonding. With respect to air plasma bombardment, a low-vacuum plasma cleaner (Harrick PDC-001) with an input power of 200 W and a maximum RF power of 30 W was used. Plasma bombardment was performed on the faying face under 0.25 Torr, which was much higher than the vacuum pressure in the plasma process for surface activation $(10^{-3}-10^{-4})$.^(14,15) Plasma treatment time was set as 1, 3, and 5 min. After plasma treatment, the samples were then stored in ambient air for hours.

The surface roughness and hardness of the samples before and after plasma exposure were measured using an atomic force microscope (AFM; Bruker Dimension Icon, tapping mode) and a nanoindenter (Hysitron TriboLab), respectively. AFM was adopted to obtain the average surface roughness of the bonding face, R_a . The scanning range was 500 × 500 nm². Nanoindentation was performed to measure the hardness and Young's modulus along the through-thickness direction. The indenter was pressed into the specimen up to 100 nm with a penetration rate of 5 nm/s. The load–displacement data obtained were analyzed using the method of Oliver and Pharr to determine the hardness and elastic modulus as functions of the displacement of the indenter.⁽¹⁶⁾ In this study, the mechanical properties in the subsurface region, about 10 nm underneath the surface, were recorded. Each datum was the average of at least 10 tests. In addition, the residual stress on the bonding face before and after air plasma bombardment was also measured. The principles of X-ray diffraction (XRD) stress measurement are described elsewhere.⁽¹⁷⁾ Grazing incidence XRD (incidence angle of 3°) was with graphite monochromatic Cu Ka radiation at a scanning rate of 1°/min with 2 θ from 40 to 80. The Cu(111) diffraction peak was selected for investigation.

A vapor-assisted bonding system was equipped with a formic acid vapor generation unit and a treatment/bonding chamber with low vacuum. Formic acid vapor was generated by importing N_2 gas into a formic acid solution. A mixed gas of formic acid vapor and N_2 was dispensed by a spray nozzle filled with Pt catalysts isothermally heated at 200 °C, which was placed between two bonding substrates. The distance between the nozzle and the two bonding pieces was about 6 mm. Thermal compression was performed at 230 or 250 °C for 5 min under nitrogen atmosphere. Bonding pressure was maintained at 10 MPa.

The shear strength was measured by breaking the bond between the joint specimens with a shear tool with a shearing rate of 0.2 mm/min. A bond tester (Cometech Testing Machines QC-506M1) was used. The shear strength was determined to be the maximum load divided by the area of Cu patterns. Each datum was the average of at least five tests. The organic residues on the fracture surface were detected using a micro Raman spectrometer.

3. Results and Discussion

The AFM images and quantitative surface roughness data given in Fig. 1 reveal that air plasma bombardment resulted in a slightly rugged surface. R_a increased from 0.8 nm (as-



Fig. 1. (Color online) AFM images of plasma-bombarded Cu films: (a) 0, (b) 1, (c) 3, and (d) 5 min, and (e) quantitative surface average roughness data.

received condition) to 2.0 nm (exposure for 1 min), 2.1 nm (exposure for 3 min), and then 2.3 nm (exposure for 5 min). Nanoindentation was conducted on the Cu surface to analyze the mechanical properties of the pretreated Cu. Figure 2 indicates the average hardness as well as Young's modulus obtained from 10 nm underneath the surface. Air plasma exposure can effectively harden the sample surface subjected to a short exposure period. Plasma irradiation can strongly increase the hardness from 2.9 to 5.9 GPa (1 min plasma exposure) and then to the maximum value of 6.4 GPa (3 min plasma exposure). However, an extended plasma exposure to 5 min resulted in a decreased hardness to 4.5 GPa. Likewise, with increasing exposure time, the Young's modulus changed from 140.4 (as-received state) to 192 (plasma exposure for 1 min), 208 (3 min), and 153 GPa (5 min). It can be deduced that plasma bombardment altered the surface mechanical features due to lattice defects and heat generation.^(18,19) The former causes work hardening (referred to as the 1- or 3-min-plasma-bombarded samples), and the latter leads to recovery (referred to as the 5-min-plasma-bombarded samples).

Thermal compression bonding can successfully join two Cu films at 250 °C or even below. Figure 3 shows the cross-sectional joint structure machined by focused ion beam of the sample subjected to 1 min air plasma bombardment, formic vapor cleaning, and then thermal compression bonding at 250 °C, depicting a firm bonding between two fayed Cu films. The bonded samples were fractured by shear test and the maximum shear strengths were recorded. It was found that all the fracturing occurred at the bonding interface. As given in Fig. 4, with a prolonged plasma treatment, the joint strengths bonded at 250 °C were 18.5 (as-received), 30.2 (1 min bombardment), 31.7 (3 min bombardment), and 28.3 MPa (5 min bombardment). When the bonding temperature was decreased from 250 to 230 °C, the joint strength was clearly deteriorated. They were 8.7 (as-received), 17.3 (1 min bombardment), 18.8 MPa (3 min bombardment), and 14.5 (5 min bombardment).

The aforementioned results reveal that subsurface hardness has a positive relationship with joint strength. On the other hand, the highest roughness value was only 2.3 nm; therefore, the influence of surface roughness was not that pronounced. Plasma bombardment can clean the sample surface and activate the surface by forming broken bonds. However, it is considered



Fig. 2. Surface hardness and Young's modulus obtained from 10 nm underneath the sample surface subjected to different periods of plasma treatment.



Fig. 3. Cross-sectional microstructure of the bonded sample (1 min air plasma bombardment).



Fig. 4. Shear strength of sputtered Cu joints subjected to different air plasma pretreating, formic acid treatment, and thermal compression bonding at 230 °C/250 °C under 10 MPa for 5 min.

that, in this case, the improvement in hydrophilicity would also be suppressed by prolonged storage in air ambient (hours in this study) or acid treatments prior to bonding.⁽²⁰⁾

Figure 5 shows the residual stress calculated from the XRD results and the comparison of Cu self-diffusivity at as-received and 3-min-plasma-exposed states, respectively. The diffusivity estimation under stress conditions can be obtained using the equation below:⁽²¹⁾

$$\ln(\frac{D}{D_0}) = (\sigma - \sigma_0) \frac{V^*}{kT},$$

where D is the diffusivity of Cu atoms under the set conditions, D_0 denotes the diffusivity of Cu at the as-received state, V^* is the atomic volume of fcc-Cu, 1.3×10^{-27} m³, and σ and σ_0 represent the stresses before and after pretreatments, respectively. It was found that the residual stresses under the as-received conditions were under the tensile state. Air plasma bombardment brought about a raised compressive stress component. If the samples were exposed to air plasma for 3 min, the tensile stress was reduced from 146 to 97 MPa. Based on the relative diffusivity data, a suppressed tensile stress gave rise to an increased D/D_0 . With air plasma exposure for 3 min, the self-diffusivity is about 1.53 times greater than those of the as-received samples.

In conjunction with our previous data related to the plasma bombardment effects on thermal compression bonding between two Cu blocks (Fig. 6), a relationship between residual stresses and joint strength can be derived. The positive contribution by a compressive stress component on direct Cu bonding can be demonstrated again.

The examination of the fracture surface using Raman spectrometers is illustrated in Fig. 7. The effect of Pt catalysis for formic acid vapor on the residual organics could be realized. The signals of v(C=O), $v_{as}CO_2^-$, $v_sCO_2^-$, and Cu(OH)₃ on the fractured surface for those subjected to formic acid vapor treatment catalyzed by Pt prior to bonding were much weaker than those with uncatalyzed vapor treatment. This leads us to believe that Pt can accelerate the dissociation of formic acid, which may be somehow beneficial for the long-term reliability of the directly bonded Cu joints.





Fig. 5. (Color online) Estimated residual stress and diffusion coefficient of Cu atoms before and after air plasma bombardment.

Fig. 6. Relationship between residual stress on faying surface and joint strength (positive stress value denotes compressive stresses, while negative value represents tensile ones).



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Raman spectra obtained from the fractured surface of sheared Cu-to-Cu joints subsequent to formic acid Fig. 7. vapor pretreatment (a) with and (b) without Pt catalysis.

4. Conclusions

In this study, we propose a concept to improve direct Cu bonding through the increase in compressive residual stresses on the faying face using air plasma pretreatments. It is suggested that instead of surface activation, plasma-induced compressive stresses, which can be verified by XRD and nanoindentation responses, contributed to an effective interdiffusion and thus robust Cu-to-Cu bonding. A joint strength higher than 30 MPa can be obtained when thermal compression bonding was performed at 250 °C for 5 min under a bonding load of 10 MPa in N₂. Formic acid vapor catalyzed by Pt was developed for an efficient on-line deoxidation prior to thermal compression bonding. It was verified that Pt catalysis can certainly diminish organic residues.

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