

Effects of Oxygen Flow Rate on Optoelectronic Characteristics of SnO₂ Thin Films for Application to Optical Sensors

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In this study, tin oxide (SnO₂) thin films were deposited on glass substrates by RF magnetron sputtering, with process parameters optimized by adjusting the oxygen flow rate. Subsequently, thermal treatment was performed at various temperatures in a vacuum environment. The thickness, electrical conductivity, optical transmittance, and figure of merit (FOM) of the films were analyzed. The results indicate that the films deposited with an oxygen flow rate of 5 sccm, a sputtering power of 80 W, and a deposition time of 30 min, followed by annealing at 200 °C, exhibited the highest performance. Specifically, the films achieved a minimum resistivity of $3.19 \times 10^{-2} \Omega\text{-cm}$ and an optimal FOM of $7.43 \times 10^{-5} \Omega^{-1}$. These findings demonstrate excellent optoelectronic properties, suggesting high potential for applications in optoelectronic sensor materials.

1. Introduction

Thin films represent a unique state of matter extensively utilized in emerging devices such as microelectronics, sensors, solar cells, and liquid crystal displays (LCDs).^(1–4) With the rapid advancement of thin-film science, various preparation techniques have been developed, ranging from vacuum evaporation to ion plating, sputtering, and chemical vapor deposition (CVD).^(5–9)

Transparent conducting oxides (TCOs) are critical optoelectronic materials that combine excellent electrical conductivity with high optical transparency, making them widely applicable in solar cells, flat-panel displays, thin-film transistors (TFTs), and organic light-emitting diodes (OLEDs).⁽³⁾ TCO films exhibit high carrier concentration and low resistivity while maintaining high transparency in the visible light spectrum. To achieve high transmittance, materials with a bandgap greater than 3 eV, such as In₂O₃, TiO₂, ZnO, SnO₂, and CdO, are typically selected.^(10–14) These materials exhibit semiconducting characteristics while maintaining optical clarity, further expanding their utility in optoelectronics. While indium tin oxide (ITO) is currently the industry standard, its widespread application is limited by the scarcity and high cost of indium, as well as its relative instability in harsh environments. Consequently, tin oxide (SnO₂) has emerged as a highly promising alternative. As an n-type semiconductor with a wide

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direct bandgap ($E_g \approx 3.6$ eV), SnO₂ offers several distinct advantages, for example, abundance and low cost, chemical and thermal stability, and mechanical durability.^(15–18)

The optoelectronic performance of SnO₂ thin films is highly sensitive to their stoichiometric composition and crystal structure, which are governed by deposition parameters. Among these, the oxygen flow rate during sputtering plays a critical role. Oxygen vacancies act as shallow donors that contribute to electrical conductivity; however, excessive oxygen can lead to a decrease in carrier concentration while improving optical transparency. Therefore, identifying the optimal oxygen flow rate is essential to achieving the best balance between electrical and optical properties, that is, the figure of merit (FOM).⁽¹⁹⁾

Despite numerous studies on SnO₂-based materials, achieving a synergistic optimization of electrical conductivity and optical clarity remains a challenge owing to the complex interplay between oxygen vacancies and structural defects. The primary objective of this study is to systematically investigate the effects of oxygen flow rate and post-annealing temperature on the optoelectronic characteristics of SnO₂ thin films deposited by RF magnetron sputtering.

We hypothesize that the precise modulation of the oxygen-to-argon ratio during deposition can effectively tailor the density of oxygen vacancies, which act as shallow donors, thereby enhancing the carrier concentration and reducing resistivity without compromising the optical bandgap. Furthermore, it is anticipated that subsequent thermal treatment will improve the crystallinity and surface morphology, leading to a superior FOM. By analyzing the correlation between deposition parameters and the resulting film properties—including thickness, resistivity, Hall mobility, and visible-light transmittance—in this work, we aim to identify the optimal processing window for high-performance SnO₂ films suitable for advanced optoelectronic sensor applications.

2. Experimental Procedure

The SnO₂ target used in this experiment was purchased from Summit International Development Co., Ltd., and Corning glass was employed as the substrate. The substrate was sequentially cleaned in deionized (DI) water, isopropyl alcohol (IPA), and acetone, followed by drying in an oven at 90 °C. Sputtering was conducted at a power of 80 W and a working pressure of 6 mTorr with various oxygen flow rates. Post-deposition thermal treatment was performed at temperatures ranging from 200 to 500 °C. A surface profilometer (alpha-step) was used to measure the thickness of the films. The total transmittance in the visible range (380–780 nm) was measured using a UV–Vis spectrophotometer. The resistivity, carrier concentration, and Hall mobility were measured at room temperature using a Hall effect measurement system (van der Pauw method).

3. Results and Discussion

3.1 Film thickness

Film thickness was measured using a surface profilometer (alpha-step). As shown in Table 1, the thin films have the same thickness even with different O₂ flow rates.⁽²⁰⁾

Table 1
Thickness of the thin films.

Sputtering power (W)	Gas flow (sccm)	Working pressure (mtorr)	Sputtering time (min)	Thickness (nm)
80	Ar: 30, O ₂ : 0	6	30	60.82
80	Ar: 30, O ₂ : 1	6	30	62.16
80	Ar: 30, O ₂ : 5	6	30	61.97
80	Ar: 30, O ₂ : 10	6	30	62.50

3.2 Analysis of electrical properties

Figures 1–4 present the results of the electrical analyses of SnO₂ films under different annealing temperatures and O₂ flow rates. The lowest resistivity of $3.19 \times 10^{-2} \Omega \cdot \text{cm}$ was observed at an O₂ flow rate of 5 sccm. The literature suggests that introducing an appropriate amount of O₂ during sputtering results in a higher concentration of oxygen vacancies, which increases carrier concentration and conductivity. However, excessive O₂ flow causes oxygen to fill these vacancies, reducing carrier concentration and increasing resistivity. Furthermore, as the annealing temperature increases, the oxidation of SnO₂ becomes more complete, further reducing oxygen vacancies and leading to higher resistivity.

The electrical resistivity (ρ) of the SnO₂ thin films is governed by the relationship $\rho = 1/(ne\mu)$, where n is the carrier concentration, e is the elementary charge, and μ is the Hall mobility. The experimental results in Figs. 1–4 reveal that the variations in resistivity are a result of the competing effects of these two parameters under different processing conditions. At lower oxygen flow rates (e.g., 5 sccm), a higher density of oxygen vacancies is maintained, resulting in a high carrier concentration. Thermal treatment promotes grain growth and improves the crystallinity of the SnO₂ lattice. Larger grains reduce the density of grain boundaries, thereby decreasing the potential barriers that impede carrier transport and enhancing mobility. The minimum resistivity of $3.19 \times 10^{-2} \Omega \cdot \text{cm}$ achieved at 200 °C with 5 sccm oxygen flow represents the point where the product of n and μ is maximized. At temperatures exceeding 300 °C, the resistivity begins to rise again. This is attributed to the “over-oxidation” of the film, where further reduction in carrier concentration (owing to the loss of oxygen vacancies) outweighs the minor gains in mobility through improved crystallinity.

3.3 Transmittance analysis

Regarding transmittance (Figs. 5–8), lower O₂ flow rates result in higher concentrations of SnO₂ phases or oxygen vacancy defects, which reduces transparency. As the O₂ flow rate increases, these defects are minimized, significantly enhancing transmittance. The optical bandgap E_g can also represent film transparency. Materials with $E_g > 3 \text{ eV}$ are classified as wide-bandgap semiconductors. The bandgap is calculated using the Tauc plot formula:^(21,22)

$$(\alpha hv)^2 = A(hv - E_g), \quad (1)$$

where A is a constant, α is the absorption coefficient, and hv is the incident photon energy. Overall, the four groups of samples showed an increasing trend in bandgap within the annealing

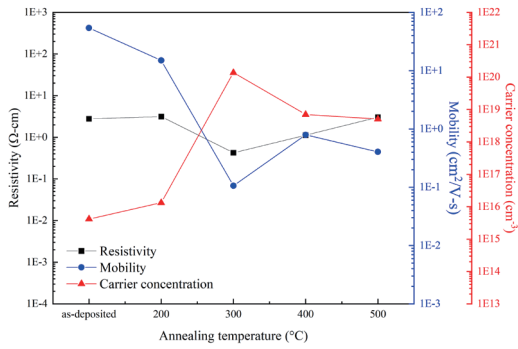


Fig. 1. (Color online) Electrical properties of SnO₂ thin films deposited at zero oxygen flow rate under different annealing temperatures.

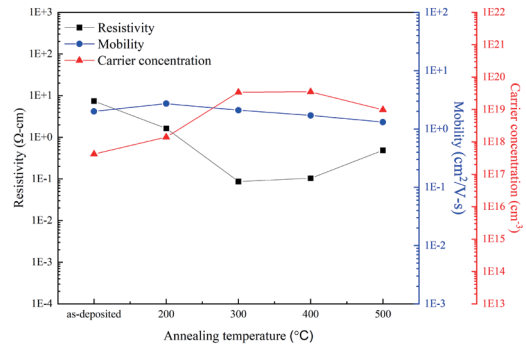


Fig. 2. (Color online) Electrical properties of SnO₂ thin films deposited at 1 sccm oxygen flow rate under different annealing temperatures.

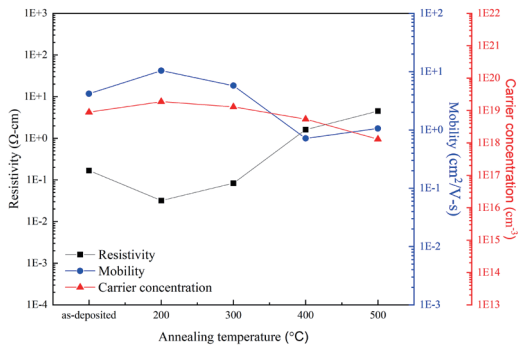


Fig. 3. (Color online) Electrical properties of SnO₂ thin films deposited at 5 sccm oxygen flow rate under different annealing temperatures.

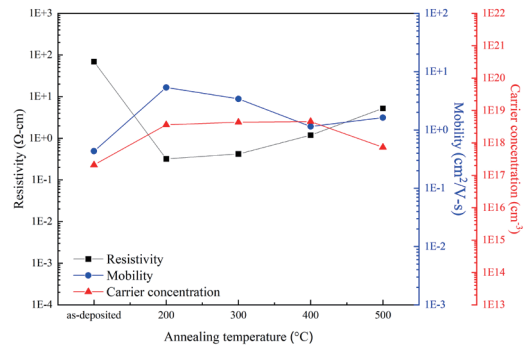


Fig. 4. (Color online) Electrical properties of SnO₂ thin films deposited at 10 sccm oxygen flow rate under different annealing temperatures.

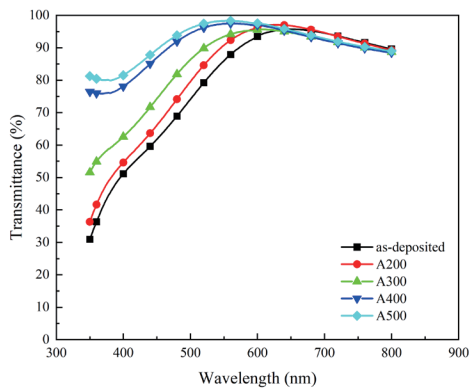


Fig. 5. (Color online) Optical transmittance of SnO₂ thin films deposited at zero oxygen flow rate under different annealing temperatures.

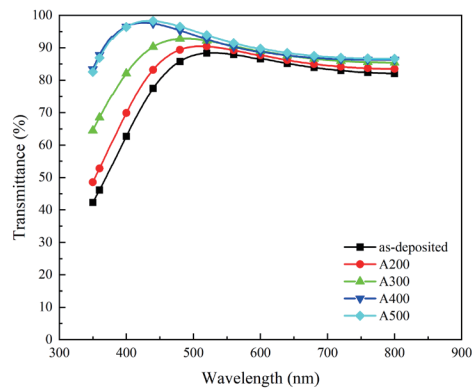


Fig. 6. (Color online) Optical transmittance of SnO₂ thin films deposited at 1 sccm oxygen flow rate under different annealing temperatures.

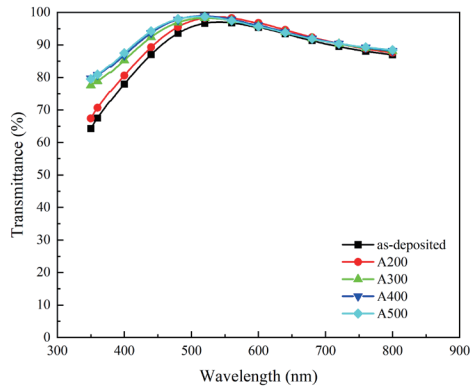


Fig. 7. (Color online) Optical transmittance of SnO₂ thin films deposited at 5 sccm oxygen flow rate under different annealing temperatures.

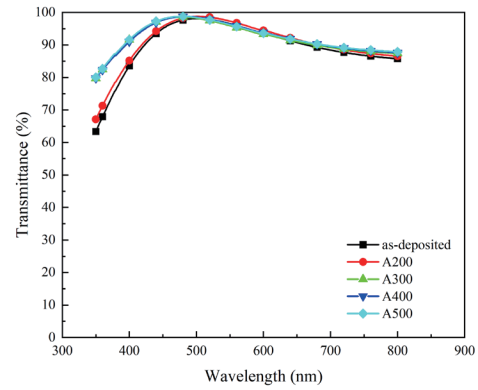


Fig. 8. (Color online) Optical transmittance of SnO₂ thin films deposited at 10 sccm oxygen flow rate under different annealing temperatures.

Table 2

Energy gap (E_g) of thin films deposited with various O₂ flow rates.

Annealing temperature (°C)	Energy Gap (eV)			
	Ar: 30	O ₂ : 1	O ₂ : 5	O ₂ : 10
as-deposited	3.06	3.1	3.59	3.53
200	3.21	3.24	3.62	3.57
300	3.52	3.5	3.71	3.64
400	3.8	3.68	3.73	3.66
500	3.84	3.68	3.73	3.67

temperature range of 200–500 °C (Table 2). All measured bandgap values were above 3 eV, consistent with typical wide-bandgap semiconductor characteristics.

3.4 FOM

The performance of TCO films is often evaluated using the FOM, defined as the ratio of transmittance to resistance:^(23–25)

$$\Phi_{TC} = \frac{T_{av}^{10}}{R_{sh}}, \quad (2)$$

where T_{av} is the average optical transmittance and R_{sh} is the sheet resistance expressed in Ω^{-1} . Since the FOM is proportional to the tenth power of transmittance, transmittance has a dominant impact. An optimal FOM requires both high transmittance and low resistivity. As shown in Fig. 9, the highest FOM was achieved at an O₂ flow rate of 5 sccm and an annealing temperature of 200 °C.

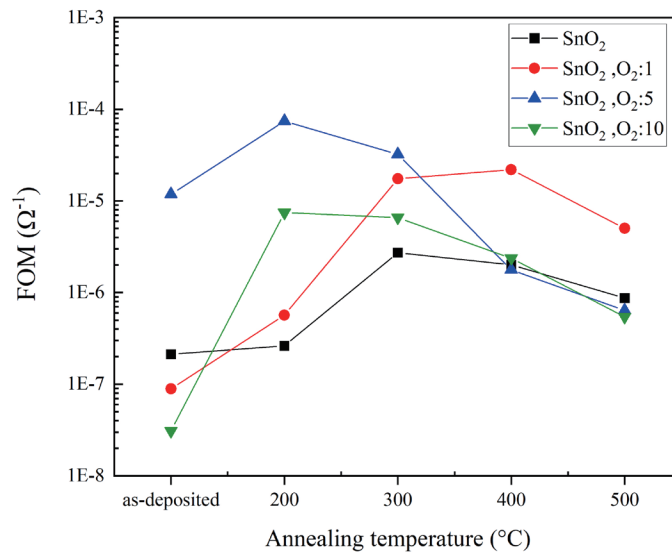


Fig. 9. (Color online) FOM of SnO₂ thin films annealed at different temperatures.

4. Conclusions

1. The thickness of SnO₂ thin films remained relatively constant regardless of O₂ flow rate or annealing temperature.
2. The minimum resistivity of $3.19 \times 10^{-2} \Omega\text{-cm}$ was obtained with 5 sccm O₂ flow and 200 °C annealing.
3. All samples exhibited average transmittance exceeding 80% and bandgap values above 3 eV.
4. The optimal FOM of $7.43 \times 10^{-5} \Omega^{-1}$ was achieved under the 5 sccm O₂ flow and annealing temperature of 200 °C, primarily owing to the superior electrical conductivity of the SnO₂ thin films.

In this study, SnO₂ transparent conductive thin films were successfully deposited by RF magnetron sputtering, and the effects of the oxygen flow rate on their physical properties were systematically investigated.

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References

- 1 T. H. Chen and C. L. Yang: Opt. Quant. Electron. **48** (2016) 533. <https://doi.org/10.1007/s11082-016-0808-3>
- 2 J. Schmitz: Surf. Coat. Technol. **343** (2018) 83. <https://doi.org/10.1016/j.surfcoat.2017.11.013>
- 3 W. N. Wang, T. D. Chang, and T. H. Chen: Sens. Mater. **36** (2024) 3757. <https://doi.org/10.18494/SAM5049>
- 4 A. Zeumault, J. E. Mendez, and J. Brewer: J. Soc. Inf. Disp. **32** (2024) 121. <https://doi.org/10.1002/jsid.1274>
- 5 C. H. Yu, S. T. Wicaksono, S. T. Wang, and T. H. Chen: Sens. Mater. **37** (2025) 1825. <https://doi.org/10.18494/SAM5316>
- 6 T. H. Chen, C. C. Chiang, and T. Y. Chen: Microsyst. Technol. **32** (2017) 1687. <https://doi.org/10.1007/s00542-015-2689-y>
- 7 M. Purica, E. Budianu, E. Rusu, M Danila, and R. Gavrila: Thin Solid Films **403–404** (2002) 485. [https://doi.org/10.1016/S0040-6090\(01\)01544-9](https://doi.org/10.1016/S0040-6090(01)01544-9)
- 8 H. Pedersen, S. T. Barry, and J. Sundqvist: J. Vac. Sci. Technol. A **39** (2021) 051001. <https://doi.org/10.1116/6.0001125>
- 9 J. F. Li, Z. Liu, Y. Zhang, B. Jia, W. Xu, X. Liu, L. Ji, A. Wang, C. Sun, and H. Li: Ceram. Int. **50** (2024) 43032. <https://doi.org/10.1016/j.ceramint.2024.08.154>
- 10 C. F. Liu, T. H. Chen, and Y. S. Huang: Sen. Mater. **32** (2020) 2321. <https://doi.org/10.18494/SAM.2020.2867>
- 11 T. H. Chen, M. W. Wang, C. L. Yang, and Y. S. Huang: Microsyst. Technol. **25** (2019) 2109. <https://doi.org/10.1007/s00542-019-04360-z>
- 12 M. Y. Yen, T. H. Chen, P. H. Lai, S. L. Tu, and Y. H. Shen: Sen. Mater. **33** (2021) 3941. <https://doi.org/10.18494/SAM.2021.3706>
- 13 A. M. Mostafa and A. A. Menazea: Radiat. Phys. Chem. **176** (2020) 109020. <https://doi.org/10.1016/j.radphyschem.2020.109020>
- 14 C. F. Liu, S. C. Shi, T. H. Chen, and G. L. Guo: Sens. Mater. **34** (2022) 4127. <https://doi.org/10.18494/SAM4133>
- 15 A. K. Gangwar, R. Godiwal, J. Jaiswal, V. Baloria, P. Pal, G. Gupta, and P. Singh: Vacuum **177** (2020) 109353. <https://doi.org/10.1016/j.vacuum.2020.109353>
- 16 A. Kania, M. M. Szindler, M. Szindler, Z. Brytan, and W. Łoński: Materials **17** (2024) 3348. <https://doi.org/10.3390/ma17133348>
- 17 Y. Xu, L. Zheng, C. Yang, W. Zheng, X. Liu, and J. Zhang: ACS Appl. Mater. Interfaces **12** (2020) 20704. <https://doi.org/10.1021/acsami.0c04398>
- 18 Y. Jang, H. Lee, and K. Char: AIP Advances **10** (2020) 035011. <https://doi.org/10.1063/1.5143468>
- 19 R. Ramarajan, N. Purushothamreddy, R. K. Dileep, M. Kovendhan, G. Veerappan, K. Thangaraju, and D. P. Joseph: Solar Energy **211** (2020) 547. <https://doi.org/10.1016/j.solener.2020.09.042>
- 20 Y. Tao, B. Zhu, Y. Yang, J. Wu, and X. Shi: Mater. Chem. Phys. **250** (2020) 123129. <https://doi.org/10.1016/j.matchemphys.2020.123129>
- 21 T. Minami, H. Nanto, and S. Takata: J. Appl. Phys. **24** (1985) L605. <https://doi.org/10.1143/JJAP.24.L605>
- 22 W. N. Wang, T. D. Chang, and T. H. Chen: Sens. Mater. **36** (2024) 3757. <https://doi.org/10.18494/SAM5049>
- 23 D. Kim: Mater. Lett. **64** (2010) 668. <https://doi.org/10.1016/j.matlet.2009.12.032>
- 24 C. Y. Lin, T. H. Chen, S. L. Tu, Y. H. Shen, and J. T. Huang: Opt Quant Electron **50** (2018) 169. <https://doi.org/10.1007/s11082-018-1430-3>
- 25 C. H. Yu, S. T. Wicaksono, S. T. Wang, and T. H. Chen: Sens. Mater. **37** (2025) 1825. <https://doi.org/10.18494/SAM5316>

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